Growth of Pure β-Si₃N₄ Nanorods from the Synergic Nitrogen Sources

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Pure β -Si₃N₄ nanorods with of 30–60 nm and length of 180–300 nm were synthesized from a mild benzene-thermal route at 450 °C, starting from SiCl₄ and the synergic nitrogen sources of NaNH₂ and NH₄Cl. The obtained sample was characterized by XRD, TEM and XPS. The one-dimensional growth mechanism is assumed to be a vapor-liquid-solid (VLS) process.

Silicon nitride (Si₃N₄) is used in a variety of important technological applications. The high fracture toughness, hardness and wear resistance of Si₃N₄-based ceramics are exploited in cutting tools and antifriction bearings.¹ In addition, Si₃N₄ is used as an insulating, masking and passivating materials in electronic applications.² Conventional synthetic methods for Si₃N₄ powder often include direct nitridation, carbothermal reduction, a silicon-sulfur-nitrogen reaction between SiS₂ and NH₃, vapor phase reactions, and thermal decomposition.^{3–9} These synthetic techniques usually require heating at elevated temperature for long periods of time, high post-treatment temperature (>1200 °C) and high production cost due to their slow growth rate. Self-propagating high-temperature synthesis (SHS) has also been applied successfully to silicon nitridation in nitrogen atmosphere, 10-13 in which an igniting agent (i.e. Ti + C) and the necessary additive (i.e. NH₄F) were used. Recently, Hu et al. employed the reaction of SiCl₄ with NaN₃ at about 670 °C under a pressure of about 45 MPa,¹⁴ in which SiCl₄ acted as both the reagent and the solvent, but only the mixture of α - and β phases of Si₃N₄ nanopowder was obtained.

Very recently, a lot of work on the preparation of one-dimensional structures has been done and important progress has been made. Nanotubes and nanowires exhibit a wide range of electronic and optical properties, which depend sensitively on their sizes and shapes.^{15–17} However, there is relatively little work on the preparation of β -Si₃N₄ nanorods.

Herein, we described the use of the reaction among SiCl₄, NaNH₂ and NH₄Cl in benzene at 450 °C in autoclave to prepare β -Si₃N₄ nanorods. In this work, we successfully employed the synergic nitrogen sources of NaNH₂ and NH₄Cl and obtained the pure β -Si₃N₄ nanorods.

In a typical process, 36 mL SiCl_4 , 0.96 g NaNH_2 and $0.32 \text{ g NH}_4\text{Cl}$ were placed in a titanium alloy autoclave of about 50 mL capacity, and then 9 mL benzene was added till up to 90% volume. Excess SiCl}4 was used to adjust the pressure in the process and the reaction temperature, thereby guaranteed the thorough completion of the reaction. After deaerated with bubbling with nitrogen gas for 30 min, the autoclave was steeled and maintained at 450 °C for 6 h, then cooled to room temperature on standing. The product was filtered out and washed with benzene to remove the unreacted SiCl_4 followed by dilute HF and distilled water to remove NaCl and other impurities. Final-

ly, the gray-white product was dried in vacuum at 60 °C for 3 h.

The XRD¹⁸ pattern corresponding to powders with random orientation is shown in Figure 1. The peaks in Figure 1 can be indexed as β -Si₃N₄ with a hexagonal cell with lattice parameters: a = 7.693 Å, c = 2.926 Å, which are consistent with the literature values (JCPDS cards No. 33-1160: a = 7.6044 Å, c = 2.9075 Å). The intensity of (111) peak increases while those of (110), (210), (101) peaks decrease, indicating the anisotropic growth of the β -Si₃N₄ crystals.

Figure 2 shows the TEM¹⁸ micrograph of the sample as well as the corresponding selected area electron diffraction pattern. The TEM image reveals β -Si₃N₄ nanorods with diameters of 30–60 nm and length of 180–300 nm (Figure 2a), and the SAED pattern clearly shows the single crystalline nature of the β -Si₃N₄ nanorods and that the nanorod axis is close to the [001] direction (Figure 2b). From this and the XRD data, it can be concluded that the nanorod axis is indeed [001].

The XPS¹⁸ spectra (Figure 3) show the composition of β -Si₃N₄ nanorods. The binding energy of N_{1s} and Si_{2p} are 397.6 eV and 101.7 eV, as shown in Figures 3a and 3b, respectively, and the quantification of peaks gives a Si:N ratio of 3.2:4. No other impurities peaks were observed.



Figure 1. XRD pattern of the asprepared β -Si₃N₄ nanocrystals.



Figure 2. (a) TEM image of β -Si₃N₄ nanorods and (b) the corresponding selected-area electron pattern (SAED).



Figure 3. XPS spectra of β -Si₃N₄ nanorods (a) Si 2p region and (b) N 1s region.

Our approach to β -Si₃N₄ nanorods is essentially based on the solvothermal reaction among SiCl₄, Na₃N and NH₃. It is well-known that NaNH₂ would decompose to Na₃N by loss of ammonia on heating¹⁹ and NH₄Cl can also decompose to NH₃ on heating, as described in the following Eqs. 2 and 1. The renascent Na₃N and NH₃ can be regarded as the intermediate reactants and immediately reacted with SiCl₄ to form β -Si₃N₄ nanocrystals, as represented in the Eq. 3. According to free energy calculation, the reaction between SiCl₄ and the synergic nitrogen source of NaNH₂ and NH₄Cl, described as the total equation, is thermodynamically driven to the right side and highly exothermic $(\Delta G_{\rm f}^{\rm o} = -696.36 \, \rm kcal \cdot mol^{-1}, \Delta H_{\rm f}^{\rm o} =$ -358.84 kcal·mol⁻¹).²⁰ In the comparison experiments, only amorphous Si₃N₄ was obtained at 450 °C, even at 500 °C, if either single nitrogen source NaNH2 or NH4Cl was used. It is the synergic effect resulting from the nitrogen sources that makes β -Si₃N₄ to crystallize at rather lower temperature. The reactions involved here are as follows:

$\begin{array}{l} \text{NH}_4\text{Cl} \longrightarrow \text{NH}_3 + \text{HCl} \\ \text{3NaNH}_2 \longrightarrow \text{Na}_3\text{N} + 2\text{NH}_3 \\ \text{3SiCl}_4 + \text{Na}_3\text{N} + 3\text{NH}_3 \longrightarrow \beta \text{-}\text{Si}_3\text{N}_4 + 3\text{Na}\text{Cl} + 9\text{HCl} \end{array}$	(1)
	(2)
	(3)

 $3SiCl_4(l) + 3NaNH_2 + NH_4Cl \xrightarrow{723\,K} \beta \text{-}Si_3N_4 + 3NaCl + 10HCl(g)$

Among the three well-known whisker-growth mechanisms, namely, solution-liquid-solid (SLS), vapor-solid (VS), and vapor-liquid-solid (VLS), the VLS mechanism is most likely to function under the present condition because this case involves the reaction in organic solvent, and uses SiCl₄ liquid reactant and the intermediate reactants including NH₃ gas reactant. In this system, SiCl₄ is much excess and is soluble in benzene. The surface of the SiCl₄–benzene mixture may have a large accommodation coefficient and is, therefore, a preferred site for the deposition of atoms from the vapor phase reactant NH₃, which can benefit the VLS nucleation for the β -Si₃N₄ nanorods. A feature of this synthesis route is the high pressure in the autoclave, coming from NH₃ and benzene, which may facilitate the VLS nucleation of the β -Si₃N₄ nanorods.

In summary, samples of 30–60 nm \times 180–300 nm β -Si₃N₄ nanorods were successfully synthesized via the reaction of SiCl₄ with the synergic nitrogen sources of NaNH₂ and NH₄Cl at 450 °C under benzene-thermal conditions. The synergic nitrogen sources play a crucial role in the formation of β -Si₃N₄ nanocrystallites. The VLS mechanism is likely to be responsible for the one-dimensional growth.

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References and Notes

- K. Komeya and M. Matsui, in "Materials Science and Technology," ed. by R. W. Cahn, P. Haasen and E. J. Kramer, Weinheim, Wiley-VCH (1994).
- 2 "Gmelin Handbook of Inorganic and Organometallic Chemistry," Si Suppl. B. 5c, Silicon Nitride in Electronic, Berlin, Springer (1991).
- 3 M. Ekelund and B. Forslund, J. Am. Ceram. Soc., **75**, 532 (1992).
- 4 T. Licko, V. Figusch, and Púchyová, *J. Eur. Ceram. Soc.*, **9**, 219 (1992).
- 5 J. Janiga, K. P. Sin, and V. Figusch, J. Eur. Ceram. Soc., 8, 153 (1991).
- 6 T. Yamada, Am. Ceram. Soc. Bull., 72, 99 (1993).
- 7 K. S. Mazdiyasni and C. M. Cooke, J. Am. Ceram. Soc., 56, 628 (1973).
- 8 P. E. D. Morgan, *Mater. Res. Soc. Symp. Proc.*, **32**, 213 (1984).
- 9 W. R. Cannon, S. C. Danforth, J. H. Flint, J. S. Haggerty, and R. A. Marra, J. Am. Ceram. Soc., 65, 324 (1982).
- 10 A. G. Merzhanov and I. P. Borovinskaya, *Combust. Sci. Technol.*, **10**, 195 (1975).
- 11 W.-C. Lee and S.-L. Chung, J. Mater. Res., 12, 805 (1997).
- 12 M. A. Rodriguez, N. S. Makhonin, J. A. Escriña, I. P. Borovinskaya, M. I. Osendi, M. F. Barba, J. E. Iglesias, and J. S. Moya, *Adv. Mater.*, 7, 745 (1995).
- 13 D. Y. Chen, B. L. Zhang, H. R. Zhuang, W. L. Li, and S. Y. Xu, *Mater. Res. Bull.*, **37**, 1481 (2002).
- 14 K. B. Tang, J. Q. Hu, Q. Y. Lu, Y. Xie, J. S. Zhu, and Y. T. Qian, Adv. Mater., 11, 653 (1999).
- 15 P. Yan, Y. Xie, Y. T. Qian, and X. M. Liu, *Chem. Commun.*, **1999**, 1293.
- 16 X. F. Duan and C. M. Lieber, J. Am. Chem. Soc., **122**, 188 (2000).
- 17 T. J. Trentler, K. M. Hickman, S. C. Goel, A. M. Viano, P. C. Gibbons, and W. E. Buhro, *Science*, **270**, 1791 (1995).
- 18 XRD measurement was carried out on a Japan Rigaku D/ max rA X-ray diffractometer equipped with graphite monochromatized high-intensity Cu-K α radiation (λ = 1.54178 Å). The TEM image and ED pattern were taken on a Hitachi Model H-800 instrument with a tungsten filament, using an accelerating voltage of 200 kV. XPS spectra were recorded on a VGESCALAB MKII X-ray photoelectron spectrometer. XPS data were collected in the constant analyzer energy (CAE) made at 20 eV with Mg K α (hv = 1253.6 eV) radiation as the excitation source.
- 19 F. A. Cotton, G. Wilkinson, C. A. Murillo, and M. Bochmann, "Advanced Inorganic Chemistry," John Wiley and Sons, Inc. (1999).
- 20 J. A. Dean, "Lange's Handbook of Chemistry," 13th ed., McGraw-Hill, New York (1985).